

REMARKS

I. Claim Amendment

New Claims 39 - 41 recite selected tempers from those recited by Claim 1.

II. 35 USC §103(a)

Claims 1, 6, 7, 10, 13-16, 18, 19, 23-25, 28, 29, 32 and 35-38 are rejected under 35 USC §103(a) as being unpatentable over Applicant's Admitted Prior Art (AAPA) in view of

Bryans et al. (US 6,973,815),  
Park (US 4,589,932) or Ohori et al. (US JP 2001-178704),  
Liu et al. (US 5,108,520) and  
Chakrabarti et al. (US2002/0150498).

A. Claims 1, 6, 7, 10, 13, 14, 18, 28, 29 and 32

Claim 1 recites bending a workpiece during cold forming and then artificially aging the bent structure. The present invention's artificial ageing after bending is in addition to other ageing that may have occurred before bending.

The Office Action in the paragraph bridging pages 2 and 3 asserts the alleged AAPA discloses a method for producing an integrated monolithic aluminum structure for a part of a wing skin or frame structure for an aircraft wherein

an AA7XXX aluminum plate with a thickness in the range of 15 to 70 mm is bent to form a predetermined shape and,

after the bending operation, the plate is machined to produce the monolithic structure,

wherein the aluminum alloy plate has been quenched and stretched (citing paragraph [0043]).

1. The Office action mischaracterized the AAPA

The Office action improperly combined two different mutually exclusive AAPA embodiments mentioned in the patent application into one single AAPA embodiment.

The AAPA discloses different methods of prior art processing.

a. Paragraph [007]

Paragraph [007], discloses a method in which:

the aluminum alloy sheet (previously artificially aged) is bent; and  
stringers or beams are attached with rivets or by welding.

Paragraph [009] explains a disadvantage of this method. "Firstly, the plate, which has been produced from an aluminum alloy which has been artificially aged as mentioned above in order to enhance the corrosion resistance, displays considerable distortion after the bending and machining operation thereby showing a vertical and horizontal distortion which makes the assembly of the aircraft fuselage or aircraft wing cumbersome since all parts need additional correction bending and measurement operations."

Also for this method page 8, paragraph [0042] explained, "When the additional components 2 are attached to the base sheet 1 and when the whole structure is finished after the machining and riveting or welding step, a horizontal distortion  $d_1$  and/or vertical distortion  $d_2$  usually results from stress relief from the pre-curved plate or sheet which has been bent before additional components 2 are connected to the base sheet or before components 2 are machined from a plate product with a corresponding thickness." This stress relieving is different from aging. Stress relieving involves heating a product for a short time period at low temperatures. Aging takes longer and is done in a controlled manner to achieve desired strength and corrosion resistance. In particular, artificial aging is performed at higher temperatures than the stress relieving mentioned for this method of the AAPA.

b. Paragraph [008]

Paragraph [008] discloses another method in which:

a plate (previously artificially aged) which has a thickness equal to  
or greater than the thickness of the sheet constituting the aircraft skin and  
the height of the stringers or beams is bent; and

after bending the stringers are machined from the plate, thereby  
milling the aluminum material from between the stringers.

Paragraph [009] explains a disadvantage of this method. "Secondly, the bent and machined structure comprising sheet and stringers or beams displays residual or inner stress originating from such bending operation and resulting in regions or parts of the structure having a microstructure different from other regions with less or more internal residual stress. Those regions with an elevated level of internal residual stress tend to be more considerably susceptible to corrosion and fatigue crack propagation."

c Paragraph [0043]

Paragraph [0043] discloses another method. In contrast, to the method of paragraph [008], the method at page 9, paragraph [0043] does not include bending. As seen in Fig. 3a, the monolithic structure is "shaped" solely by mechanical milling or machining to convert an aluminum alloy block to predetermined shaped structure 5.

d. The improper combination

The Office action has combined two different AAPA's mentioned in the patent to one single AAPA. In particular, the Office action combined different features from the method of paragraph [008] and the method at page 9, paragraph [0043]. It is respectfully submitted this is improper.

In the method of paragraph [008] of the patent specification, mention is made of a known method of producing an aircraft fuselage skin from an aluminum alloy plate having a thickness in the range of 15-70 mm. The plate is bent and, after bending, the stringers are machined from the plate. No mention is made of the type of alloy used in this AAPA, or of the temper of the plate before bending.

In contrast, paragraph [043] of the patent specification relates to a different type of prior art, in which a very thick plate is produced and then directly machined down to obtain a predetermined shaped structure, see Fig. 3a. The word "shaped" in the paragraph is not used in the sense of "bending", paragraph [043] explains the shaping step is a mechanical milling or machining step. Thus, paragraph [043] describes an entirely different type of prior art. It is not proper to combine the method of paragraph [008] and the method of paragraph [0043] into one disclosure.

It is respectfully submitted, of these two AAPA methods, the AAPA of paragraph [008] is more relevant to the present invention because it includes a bending step.

One of the main differences of the invention of Claim 1 over the AAPA of

paragraph [008] is that in the invention the plate is first shaped to a shaped structure having a built-in radius, then the shaped structure is heat-treated by artificial ageing to the second temper and machined to obtain the integrated monolithic aluminum structure. Claim 38 further specifies the machining is after the artificial ageing. The AAPA of paragraph [008] lacks heat-treating by artificial ageing to the second temper.

2. There is no reason to heat treat Bryans' bent material

The Office action also asserts the AAPA does not disclose the bending operation being a cold-forming of a AA7XX aluminum plate that has been brought to a temper selected from the group consisting of T4, T73, T74 and T76 and heat treating by artificially aging said shaped structure to a second temper selected from the group consisting of T6, T79, T78, T77, T76, T74, T73 or T8. However, the Office action asserts Bryans cold forms an aeronautical member from an AA7XXX aluminum alloy plate in a T7451 temper by bending. Thus, it would have been obvious to use in the AAPA an alloy plate in a T7451 temper and bend the aluminum alloy plate in the T7451 temper.

It is respectfully submitted, Bryans teaches away from heat-treating the shaped structure. Bryans, col. 5, lines 3-13 explicitly discloses the material need not be subjected to subsequent heat-treatments.

In some embodiments, a suitable material for use in the process is a 7000 series aircraft aluminum alloys (*sic*) with a heat treatment of T7451. In some embodiments, the selected material is cold formed, meaning no heat is applied to the material during the process; in some embodiments, the material is formed and processed in a state wherein it was previously heat treated. Further, in the process of the present invention, the selected material need not be subjected to subsequent heat treatments or annealing operations, and the amount of over forming and bending back is may be (*sic*) selected and controlled."

Thus, there is no reason to complicate Bryan's process by inserting heat treatment between Bryans' forming and machining steps. Bryans already has its

material in its desired temper and Bryans et al. teaches machining directly after forming.

Furthermore, Bryans, col. 8, lines 10-17, proposes its own way to manage forming and temper:

In some embodiments, there will be tensile stress on the outer skin side of the part produced by the process of the present invention, and compressive stresses on the inside integral supporting structure. This may be a result of the forming process and staying within prescribed limits of bend v. plate thickness by alloy and temper (so properties of the selected material are not comprised (*sic, compromised?*)).

Claims 40 and 41 avoid T4 temper to further distinguish over Bryans.

3. Park teaches away from TXXX alloys

The AAPA of paragraph [008] does not disclose the step of heat-treating to the second temper between shaping and machining.

Bryans does not disclose the step of heat-treating to the second temper between shaping and machining. Thus, to make up for this deficiency in the AAPA and Bryans, the Office action cites Park or Otori.

It is respectfully submitted it is improper to combine Park with the AAPA as modified to have the AA7XXX alloy of Bryans. Park relates to a completely different alloy, namely an AA6XXX series alloy (see Abstract), and very importantly, to a different use, namely to vehicular members (see column 1, line 10).

Park relates to automotive body parts produced by casting, homogenizing, hot-rolling and cold-rolling to a final gauge much less than 10 mm (see column 5, lines 24-52 and column 8, line 7, where a 0.1 inch thickness is mentioned). This sheet is solution heat treated and quenched. After quenching, the sheet may age naturally until the time the sheet is shaped to its final shape, normally a particular automotive body panel. After shaping, the part can be artificially aged to T6 temper. In particular, Park, col. 6, lines 39-55 discloses "it is generally recognized that a shaping operation can be interposed between solution heat treating and artificial aging operations to advantage

since the moderate strength and higher workability of the T4 temper facilitate such which can be followed by the strength improving operation of artificial aging to produce the T6 type temper."

However, Park only discloses this treatment for 6XXX-series alloys which have quite different properties than the 7XXX series alloys of the present invention. Park even distinguishes the invention over 7XXX-series alloys, see column 14, lines 58-62 "Equally significant is the fact that 7XXX alloys, when substituted for 6061, also include a forming penalty in that 7XXX alloys are more difficult to form and when so shaped exhibit residual stress in the frame." Thus, Park teaches away from subjecting 7XXX alloys to the shaping treatment recommended for 6XXX alloys.

Furthermore, the products of Park are not stretched before bending, as in step a) of claim 1.

Another difference between the AAPA and Park is that the sheets and extrusions of Park are too thin to be machined down. Therefore, the one skilled in the art would not consider this reference.

Finally, Park does not mention aerospace applications at all, but only vehicular and sporting applications such as ski poles and baseball bats (see col. 14, lines 10-25). Generally, the skilled person for aluminum alloys for automotive uses is not the same skilled person as for aerospace uses. Therefore, it is improper to combine Park with the AAPA, even if modified by Bryans, concerning the manufacture of aerospace members.

Claim 41 avoids T6 temper to further distinguish over the AAPA, even if combined with Bryans and Park.

4. Ogori teaches away from 7XXX alloys

As an alternative to Park, the Office action cites Ogori (JP 2000178704). However, this reference concerns extruded products of a 6000-series alloy. In contrast, the present invention relates to AA 7XXX series alloy.

AA 7XXX series alloys differ from AA 6XXX series alloys. The alloy designation in the 2XXX through 8XXX groups is determined by the alloying element (Mg<sub>2</sub>Si for 6XXX alloys) present in the greatest mean percentage (ATTACHMENT I, International

Alloy Designations and Chemical Composition Limits for Wrought Aluminum and Wrought Aluminum Alloys, The Aluminum Association, p. 18 (January 2001). In contrast, zinc, in amounts of 1 to 8 %, is the major alloying element in AA 7XXX series alloy (See p. 59 of ATTACHMENT II, ASM Specialty Handbook, Aluminum and Aluminum Alloys, J. Davis, ed., ASM International (1993)).

Also, as mentioned above, Ohori concerns extruded products. Therefore, it is evidently improper to combine this reference with the AAPA even if modified by Bryans, because a plate product according to the invention cannot be produced by extrusion.

Furthermore, Ohori relates to extrusions for motor vehicles, while the invention is concerned with aerospace products.

The Office action asserts Ohori teaches a double heat-treatment of ageing aluminum alloy before and after bending. However, the abstract available from esp@cenet-Bibiligraphic Data of the European Patent Office states the extruded shape is first aged to T1, and after bending aged to T2 condition (ATTACHMENT III, esp@cenet-Bibiligraphic Data for JP 2000178704). T1 and T2 are quite different temper conditions than T4 and T6 and T7 and essentially cannot be compared (See p.p. 29-30 of ATTACHMENT III, ASM Specialty Handbook, Aluminum and Aluminum Alloys, J. Davis, ed., ASM International (1993)).

T1 applies to product cooled from an elevated-temperature shaping process and naturally aged to a substantially stable condition. T2 applies to a product cooled from an elevated-temperature shaping process, cold-worked and naturally aged to a substantially stable condition. (ATTACHMENT II, p.p. 29-30).

T4 applies to product that has been solution heat treated and naturally aged to a substantially stable condition. T6 applies to product that has been solution heat-treated and artificially aged. T7 applies to product that has been solution heat-treated and over aged or stabilized. (ATTACHMENT II, p.p. 29-30).

5. Liu et al. and Chakrabarti et al.

Page 4 of the Office action cited Liu et al. for composition of 7xxx-series of aluminum alloys. Chakrabarti et al. at page 4 of the Office action was cited for

disclosing properties of various tempers.

It is submitted neither of these references makes up for the above-discussed deficiencies of the combination of AAPA, Bryans et al and Park or Ohori.

B. Claims 15-16, 19 and 23-25

These claims further distinguish the present invention over the references.

C. Claims 35-37

Claims 35-37 further distinguish over the references as they recite the method of manufacturing various structural parts of an aircraft.

D. Claim 38

Claim 38 further distinguishes over the references by reciting machining after artificial ageing.

E. Claims 39-41

Claim 39-41 further distinguish over the references by reciting selected tempers.

III. Conclusion

In view of the above it is respectfully submitted that all objections and rejections are overcome. Thus, a Notice of Allowance is respectfully requested.

Respectfully submitted,  
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Date: March 23, 2009

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REGISTRATION RECORD SERIES

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**International Alloy Designations  
and  
Chemical Composition Limits  
for  
Wrought Aluminum and  
Wrought Aluminum Alloys**

*Unified North American and International Registration Records*

**The Aluminum Association**

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**Revised: January 2001**

**Supersedes: July 1998**

# ATTACHMENT I, International Alloy Designations and Chemical Composition Limits for Wrought Aluminum and Wrought Aluminum Alloys, The Aluminum Association, p. 18 (January 2001)

## RECOMMENDATION INTERNATIONAL DESIGNATION SYSTEM FOR WROUGHT ALUMINUM AND WROUGHT ALUMINUM ALLOYS

This Recommendation is based on the numerical designation system for wrought aluminum and wrought aluminum alloys which was adopted in the U.S.A. in 1954, and became a national standard in 1957. This Recommendation was officially adopted by the International Signatories of the Declaration of Accord on December 15, 1970 and was editorially revised on May 1, 1974.

Designations in accordance with this Recommendation may be used by any country, but there is no obligation to use them. For use, see Appendices A and B.

A numerical designation assigned in conformance with this Recommendation should only be used to indicate an aluminum or an aluminum alloy having chemical composition limits identical to those registered with the Signatories to the Declaration of Accord on an International Alloy Designation System for Wrought Aluminum and Wrought Aluminum Alloys.

### 1. Scope

This recommendation describes a four-digit numerical system for designating wrought aluminum and wrought aluminum alloys.

### 2. Alloy Groups

The first of the four digits in the designation indicates the alloy group as follows:

Aluminum, 99.00 percent and greater	1xxx
Aluminum alloys grouped by major alloying elements <sup>1,2,3</sup>	
Copper	2xxx
Manganese	3xxx
Silicon	4xxx
Magnesium	5xxx
Magnesium and Silicon	6xxx
Zinc	7xxx
Other elements	8xxx
Unalloyed series	9xxx

### 3. 1xxx Group

The designation assigned shall be in the 1xxx group whenever the minimum aluminum content is specified as 99.00 percent and greater. In the 1xxx group, the last two of the four digits in the designation indicate the minimum aluminum percentage<sup>4</sup>. These digits are the same as the two digits to the right of the decimal point in minimum aluminum percentage when it is expressed to the nearest 0.01 percent. The second digit in the alloy designation indicates alloy modifications in impurity limits or alloying elements. If the second digit in the designation is zero, it indicates unalloyed aluminum having natural impurity limits; integers 1 through 9, which are assigned consecutively as needed, indicate special control of one or more individual impurities or alloying elements.

### 4. 2xxx-8xxx Groups

The alloy designation in the 2xxx through 8xxx groups is determined by the alloying element (Mg<sub>2</sub>Si for 6xxx alloys) present in the greatest mean percentage. If the greatest mean percentage is common to more than one alloying element, choice of group will be in order of group sequence Cu, Mn, Si, Mg, Mg<sub>2</sub>Si, Zn or Others. In the 2xxx through 8xxx alloy groups the last two of the four digits in the designation have no special significance but serve only to identify the different aluminum alloys in the group. The second digit in the alloy designation indicates the original alloy<sup>5</sup> and alloy modifications, integers 1 through 9, which are assigned consecutively, indicate alloy modifications.

### 5. Modifications

A modification of the original alloy<sup>6</sup> is limited to any one or a combination of the following:

- Change of not more than the following amounts in the arithmetic mean of the limits for an individual alloying element or combination of elements expressed as an alloying element or both:

Arithmetic Mean of Limits for Alloying Elements in Original Alloy	Maximum Change
Up through 1.0 percent	0.15
Over 1.0 through 2.0 percent	0.20
Over 2.0 through 3.0 percent	0.25
Over 3.0 through 4.0 percent	0.30
Over 4.0 through 5.0 percent	0.35
Over 5.0 through 6.0 percent	0.40
Over 6.0 percent	0.50

To determine compliance when maximum and minimum limits are specified for a combination of two or more elements in one alloy composition, the mean of such combination should be compared to the sum of the mean values of the same individual elements, or any combination thereof, in another alloy composition.

- Addition or deletion of not more than one alloying element with limits having an arithmetic mean of not more than 0.30 percent, or addition or deletion of not more than one combination of elements expressed as an alloying element with limits having a combined arithmetic mean of not more than 0.40 percent.
- Substitution of one alloying element for another element serving the same purposes.
- Change in limits for impurities expressed singly or as a combination.
- Change in limits for grain refining elements.
- Maximum iron or silicon limits of 0.12 percent and 0.10 percent, or less, respectively, reflecting high purity base metal.

An alloy should not be registered as a modification if it meets the requirements for a national variation.

### 6. National Variations

National variations of wrought aluminum and wrought aluminum alloys registered by another country in accordance with this Recommendation are identified by a serial letter after the numerical designation. The serial letters are assigned in alphabetical sequence starting with A for the first national variation registered, but omitting I, O, and Q.

A national variation has composition limits which are similar but not identical to those registered by another country, with differences such as:

- Differences in arithmetic mean of limits for an individual alloying element or combination of elements expressed as an alloying element, or both, not exceeding the following amounts:

Arithmetic Mean of Limits for Alloying Elements in Original Alloy or Modification	Maximum Difference
Up through 1.0 percent	0.15
Over 1.0 through 2.0 percent	0.20
Over 2.0 through 3.0 percent	0.25
Over 3.0 through 4.0 percent	0.30
Over 4.0 through 5.0 percent	0.35
Over 5.0 through 6.0 percent	0.40
Over 6.0 percent	0.50

To determine compliance when maximum and minimum limits are specified for a combination of two or more elements in one alloy composition, the mean of such combination should be compared to the sum of the mean values of the same individual elements, or any combination thereof, in another alloy composition.

- Substitution of one alloying element for another element serving the same purpose.
- Different limits of impurities except for low iron. Iron maximum of 0.12 percent or less, reflecting high purity base metal, should be considered an alloy modification. See 5(f).
- Different limits on grain refining elements.
- Inclusion of a minimum limit for iron or silicon, or both.

An alloy meeting these requirements should not be registered as a new alloy or alloy modification.

See footnotes on page 19

# Aluminum and Aluminum Alloys

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ATTACHMENT II - ASM Specialty Handbook, Aluminum and Aluminum Alloys, J. Davis, ed., p.p. 29-30 and 59-62, ASM International (1993))

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First printing, December 1993  
Second printing, February 1994  
Third printing, March 1996  
Fourth printing, March 1998

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Library of Congress Cataloging-in-Publication Data

Aluminum and aluminum alloys / edited by J.R. Davis;  
prepared under the direction of the ASM International Handbook Committee.  
p. cm. -- (ASM specialty handbook)  
Includes bibliographical references and index.

1. Aluminum. 2. Aluminum alloys.  
I. Davis, J.R. (Joseph R.)
  - II. ASM International. Handbook committee.
  - III. Series.
- TA480.A6A6177 1993 629.1 '86-dc20

ISBN: 0-87170-496-X

ASM International®  
Materials Park, OH 440730002

Printed in the United States of America

temper designations have already been assigned for wrought products in all alloys.

- *Hx1* applies to products that incur sufficient strain hardening after final annealing to fail to qualify as 0 temper, but not so much or so consistent an amount of strain hardening to qualify as *Hx1* temper.
- *H12* pertains in products that may acquire some strain hardening during working at elevated temperatures and for which there are mechanical property limits.
- *Patterned or Embossed Sheet*. Table 5 lists the three-digit 0 temper designations that have been assigned to patterned or embossed sheet.

### System for Heat-Treatable Alloys

The temper designation system for wrought and cast products that are strengthened by heat treatment employs the W and T designations described in the section "Basic Temper Designations" in this article. The W designation denotes an unstable temper, whereas the T designation denotes a stable temper other than F, O, or H. The T is followed by a numeral from 1 to 10, each numeral indicating a specific sequence of basic treatments. A description of how aluminum alloys are classified as heat-treatable versus non-heat-treatable can be

found in the article "General Introduction" in this volume.

**T1. Cooked from an Elevated-Temperature Shaping Process and Naturally Aged to a Substantially Stable Condition.** This designation applies to products that are not cold-worked after an elevated-temperature shaping process such as casting or extrusion and for which mechanical properties have been stabilized by room-temperature aging. It also applies to products that are flattened or straightened after cooling from the shaping process, for which the effects of the cold work imparted by flattening or straightening are not accounted for in specific property limits.

**T2. Cooled from an Elevated-Temperature Shaping Process, Cold-Worked, and Naturally Aged to a Substantially Stable Condition.** This variation refers to products that are cold-worked specifically to improve strength after cooling from a hot-working process such as rolling or extrusion and for which mechanical properties have been stabilized by room-temperature aging. It also applies to products in which the effects of cold work, imparted by flattening or straightening, are accounted for in specified property limits.

**T3, Solution Heat-Treated, Cold-Worked, and Naturally Aged to a Substantially Stable Condition.** T3 applies to products that are cold-worked specifically to improve strength after solution heat treatment and for

which mechanical properties have been stabilized by slow-temperature aging. It also applies to products in which the effects of cold work, imparted by flattening or straightening, are accounted for in specified property limits.

**T4. Solution Heat-Treated and Naturally Aged to a Substantially Stable Condition.** This signifies products that are not cold-worked after solution heat treatment and for which mechanical properties have been stabilized by room-temperature aging. If the products are flattened or straightened, the effects of the cold work imparted by flattening or straightening are not accounted for in specified property limits.

**T3, Cooled from an Elevated-Temperature Shaping Process and Artificially Aged.** TS includes products that are not cold-worked after an elevated-temperature shaping process such as casting or extrusion and for which mechanical properties have been substantially improved by precipitation heat treatment. If the products are flattened or straightened after cooling from the shaping process, the effects of the cold work imparted by flattening or straightening are not accounted for in specified property limits.

**T6, Solution Heat-Treated and Artificially Aged.** This group encompasses products that are not cold-worked after solution heat treatment and for which mechanical properties or dimensional stability, or both, have been substantially improved by precipitation heat treatment. If the products are flattened or straightened, the

Table 4 ISO equivalents of wrought Aluminum Association international alloy designations

[illegible]

Table 5 H temper designations for aluminum and aluminum alloy patterned or embossed sheet

Putnam or Hendall sheet	Range of sheet from which measured sheet was fabricated	
HE140	HE141	13
HE141	HE142	14
HE142	HE143	15
HE143	HE144	16
HE144	HE145	17
HE145	HE146	18
HE146	HE147	19
HE147	HE148	20
HE148	HE149	21
HE149	HE150	22
HE150	HE151	23
HE151	HE152	24
HE152	HE153	25
HE153	HE154	26
HE154	HE155	27
HE155	HE156	28
HE156	HE157	29
HE157	HE158	30
HE158	HE159	31
HE159	HE160	32
HE160	HE161	33
HE161	HE162	34
HE162	HE163	35
HE163	HE164	36
HE164	HE165	37
HE165	HE166	38
HE166	HE167	39
HE167	HE168	40
HE168	HE169	41
HE169	HE170	42
HE170	HE171	43
HE171	HE172	44
HE172	HE173	45
HE173	HE174	46
HE174	HE175	47
HE175	HE176	48
HE176	HE177	49
HE177	HE178	50
HE178	HE179	51
HE179	HE180	52
HE180	HE181	53
HE181	HE182	54
HE182	HE183	55
HE183	HE184	56
HE184	HE185	57
HE185	HE186	58
HE186	HE187	59
HE187	HE188	60
HE188	HE189	61
HE189	HE190	62
HE190	HE191	63
HE191	HE192	64
HE192	HE193	65
HE193	HE194	66
HE194	HE195	67
HE195	HE196	68
HE196	HE197	69
HE197	HE198	70
HE198	HE199	71
HE199	HE200	72
HE200	HE201	73
HE201	HE202	74
HE202	HE203	75
HE203	HE204	76
HE204	HE205	77
HE205	HE206	78
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HE208	HE209	81
HE209	HE210	82
HE210	HE211	83
HE211	HE212	84
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HE215	HE216	88
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HE236	HE237	109
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HE238	HE239	111
HE239	HE240	112
HE240	HE241	113
HE241	HE242	114
HE242	HE243	115
HE243	HE244	116
HE244	HE245	117
HE245	HE246	118
HE246	HE247	119
HE247	HE248	120
HE248	HE249	121
HE249	HE250	122
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HE252	HE253	125
HE253	HE254	126
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HE255	HE256	128
HE256	HE257	129
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HE258	HE259	131
HE259	HE260	132
HE260	HE261	133
HE261	HE262	134
HE262	HE263	135
HE263	HE264	136
HE264	HE265	137
HE265	HE266	138
HE266	HE267	139
HE267	HE268	140
HE268	HE269	141
HE269	HE270	142
HE270	HE271	143
HE271	HE272	

Source: Ind:

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effects of the cold work imparted by flattening or straightening are not accounted for in specified property limits.

**T7, Solution Heat-Treated and Overaged or Stabilized.** T7 applies to wrought products that have been precipitation heat-treated beyond the point of maximum strength to provide some special characteristic, such as enhanced resistance to stress-corrosion cracking or exfoliation corrosion (both of these modes of corrosion are described in the article "Corrosion Behavior" in this Volume). It applies to cast products that are artificially aged after solution heat treatment to provide dimensional and strength stability.

**T8, Solution Heat-Treated, Cold-Worked, and Artificially Aged.** This designation applies to products that are cold-worked specifically to improve strength after solution heat treatment and for which mechanical properties or dimensional stability, or both, have been substantially improved by precipitation heat treatment. The effects of cold work, including any cold work imparted by flattening or straightening, are accounted for in specified property limits.

**T9, Solution Heat-Treated, Artificially Aged, and Cold-Worked.** This grouping is comprised of products that are cold-worked specifically to improve strength after they have been precipitation heat-treated.

**T10, Cooled from an Elevated-Temperature Shaping Process, Cold-Worked, and Artificially Aged.** T10 identifies products that are cold-worked specifically to improve strength after cooling from a hot-working process such as rolling or extrusion and for which mechanical properties have been substantially improved by precipitation heat treatment. The effects of cold work, including any cold work imparted by flattening or straightening, are accounted for in specified property limits.

**Additional T Temper Variations.** When it is desirable to identify a variation of one of the ten major T tempers described above, additional digits, the first of which cannot be zero, may be added to the designation.

Specific sets of additional digits have been assigned to stress-relieved wrought products:

*Stress-Relieved by Stretching, Compressing, or Combination of Stretching and Compressing.* This designation applies to the following products when stretched to the indicated amounts after solution heat treatment or after cooling from an elevated-temperature shaping process:

Product form:	Permanent set, %
Plate	1½-3
Rod, bar, shapes, and extruded tube	1-3
Drawn tube	3½-5

- T51 applies specifically to plate, to rolled or cold-finished rod and bar, to the die or ring forgings, and to rolled rings. These products receive no further straightening after stretching.

- T51G applies to extruded rod, bar, shapes and tubing, and to drawn tubing. Products in this temper receive no further straightening after stretching.
- T51U refers to products that may receive minor straightening after stretching to comply with standard tolerances.

One variation involves stress relief by compressing:

- T52 applies to products that are stress-relieved by compressing after solution heat treatment or after cooling from a hot-working process to produce a permanent set of 1 to 5%.

The next designation is used for products that are stress-relieved by combining stretching and compressing:

- T54 applies to die forgings that are stress-relieved by reworking cold in the finish die. (These same digits—51, 52, and 54—may be added to the designation W to indicate unstable solution heat-treated and stress-relieved tempers.)

Temper designations have been assigned to wrought products heat-treated from the O or the F temper to demonstrate response to heat treatment:

- T42 means solution heat treated from the O or the F temper to demonstrate response to heat treatment and naturally aged to a substantially stable condition.
- T62 means solution heat-treated from the O or the F temper to demonstrate response to heat treatment and artificially aged.

Temper designations T42 and T62 also may be applied to wrought products heat-treated from any temper by the user when such heat treatment results in the mechanical properties applicable to these tempers.

#### System for Annealed Products

A digit following the O indicated a product in annealed condition having special characteristics. For example, for heat-treatable alloys, O1 indicates a product that has been heat-treated at approximately the same time and temperature required for solution heat treatment and then air-cooled to room temperature; this designation applies to products that are to be machined prior to solution heat treatment by the user. Mechanical property limits are not applicable.

#### Designation of Unregistered Tempers

The letter P has been assigned to denote H, T, and O temper variations that are negotiated between manufacturer and purchaser. The let-

ter P follows the temper designation that most nearly pertains. The use of this type of designation includes situations where:

- The use of the temper is sufficiently limited to preclude its registration.
- The test conditions are different from those required for registration with the Aluminum Association.
- The mechanical property limits are not established on the same basis as required for registration with the Aluminum Association.

#### Foreign Temper Designations

Unlike the agreement relating to wrought alloy designations, there is no Declaration of Accord for an international system of tempers to be registered with the Aluminum Association by foreign organizations. For the most part, the ANSI system is used, but because there is no international accord, reference to ANSI H3.1 properties and characteristics of aluminum alloy tempers registered with the Aluminum Association under ANSI 35.1 may not always reflect actual properties and characteristics associated with the particular alloy temper. In addition, temper designations may be created that are not registered with the Aluminum Association.

#### ACKNOWLEDGMENT

The information in this article is largely taken from R.E.C. Clayless, Alloy and Temper Designation Systems for Aluminum and Aluminum Alloys, Volume 2 of the *ASM Handbook* (formerly *Metals Handbook*, 10th Edition), ASM International, 1990, p. 15-28.

#### REFERENCES

1. "American National Standard Alloy and Temper Designation Systems for Aluminum," Aluminum Association, Washington, D.C., 1990
2. "Registration Record of International Alloy Designations and Chemical Composition Limits for Wrought Aluminum and Wrought Aluminum Alloys," Aluminum Association, Washington, D.C., 1991
3. *Metals and Alloys in the Unified Numbering System*, 6th ed., Society of Automotive Engineers, Warrendale, PA, 1993
4. J.G. Cernuschi and E.L. Potts, Eds., *International Metallic Materials Cross-Reference*, 4th ed., Geom Publishing, 1989
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## Wrought Products

ALUMINUM wrought products are those aluminum products that have been subjected to plastic deformation by hot working and cold working processes (such as rolling, extruding, and drawing, either singly or in combination), so as to transform cast aluminum ingots into desired product form.

The microstructural changes associated with the working and with any accompanying thermal treatments are used to control certain properties and characteristics of the worked, or wrought, product or alloy.

Typical examples of wrought products include plate or sheet (which is subsequently formed or machined into products such as aircraft or building components), household foil, extruded shapes such as storm window frames, and forged automotive and airframe components. A vast difference in the mechanical and physical properties of aluminum wrought products can be obtained through the control of the chemistry, processing, and thermal treatment.

### General Characteristics of Wrought Alloys

Aluminum alloys are commonly grouped into an alloy designation series, as described in the article "Alloy and Temper Designation Systems" in this Volume. The general characteristics of the alloy groups are described below, and the comparative corrosion and fabrication characteristics and some typical applications of the commonly used grades or alloys in each group are presented in Table 1.

**1xxx Series.** Aluminum of 99.0% or higher purity has many applications, especially in the electrical and chemical fields. These grades of aluminum are characterized by excellent corrosion resistance, high thermal and electrical conductivity, low mechanical properties, and excellent workability. Moderate increases in strength may be obtained by strain hardening. Iron and silicon are the major impurities. Typical uses include chemical equipment, reflectors, heat exchangers, electrical conductors and capacitors, packaging foil, architectural applications, and decorative trim.

**2xxx Series.** Copper is the principal alloying element in 2xxx series alloys, often with magnesium as a secondary addition. These alloys

require solution heat treatment to obtain optimum properties; in the solution heat-treated condition, mechanical properties are similar to, and sometimes exceed, those of low-carbon steel. In some instances, precipitation heat treatment (aging) is employed to further increase mechanical properties. This treatment increases yield strength, with attendant loss in elongation; its effect on tensile strength is not as great.

The alloys in the 2xxx series do not have corrosion resistance as good as that of most other aluminum alloys, and under certain conditions they may be subject to intergranular corrosion. Therefore, these alloys in the form of sheet usually are clad with a high-purity aluminum or with a magnesium-silicon alloy of the 6xxx series, which provides galvanic protection of the core material and thus greatly increases resistance to corrosion.

Alloys in the 2xxx series are particularly well suited for parts and structures requiring high strength-to-weight ratios and are commonly used to make truck and aircraft wheels, truck suspension parts, aircraft fuselage and wing skins, and structural parts and those parts requiring good strength at temperatures up to 150 °C (300 °F). Except for alloy 2219, these alloys have limited weldability, but some alloys in this series have superior machinability.

**3xxx Series.** Manganese is the major alloying element of 3xxx series alloys. These alloys generally are non-heat-treatable but have about 20% more strength than 1xxx series alloys. Because only a limited percentage of manganese (up to about 1.5%) can be effectively added to aluminum, manganese is used as a major element in only a few alloys. However, three of them—3003, 3004, and 3105—are widely used as general-purpose alloys for moderate-strength applications requiring good workability. These applications include beverage cans, cooking utensils, heat exchangers, storage tanks, awnings, furniture, highway signs, roofing, siding, and other architectural applications.

**4xxx Series.** The major alloying element in 4xxx series alloys is silicon, which can be added in sufficient quantities (up to 12%) to cause substantial lowering of the melting range without producing brittleness. For this reason, aluminum-silicon alloys are used in welding wire and as brazing alloys for joining aluminum, where a lower melting range than that of the base metal is required. Most alloys in this series are non-heat-treatable, but when used in welding heat-treatable

alloys, they will pick up some of the alloying constituents of the latter and to respond to heat treatment to a limited extent. The alloys containing appreciable amounts of silicon become dark gray to charcoal when anodic oxide finishes are applied and hence are in demand for architectural applications. Alloy 4032 has a low wear resistance, and thus it is well suited to production of forged engine pistons.

**5xxx Series.** The major alloying element in 5xxx series alloys is magnesium. When it is used as a major alloying element (with manganese), the result is a moderate-to-high-strength work-hardenable alloy. Magnesium is considerably more effective than manganese as a hardener, about 0.8% Mg being equal to 1.25% Mn, and it can be added in considerably higher quantities. Alloys in this series possess good welding characteristics and good resistance to corrosion in marine atmospheres. However, certain limitations should be placed on the amount of cold work and the safe operating temperatures permissible for the higher-magnesium alloys (over about 3.5% for operating temperatures above about 65 °C, or 150 °F) to avoid susceptibility to stress-corrosion cracking.

Uses include architectural, ornamental, and decorative trim; cars and car ends; household appliances; aircraft standards; boats and ships; cryogenic tanks; crane parts; and automotive structures.

**6xxx Series.** Alloys in the 6xxx series contain silicon and magnesium approximately in the proportions required for formation of magnesium silicide ( $Mg_2Si$ ), thus making them heat-treatable. Although not as strong as most 2xxx and 7xxx alloys, 6xxx series alloys have good formability, weldability, machinability, and corrosion resistance, with median strength. Alloys in this heat-treatable group may be formed in the T4 temper (solution heat-treated but not precipitation heat-treated) and strengthened after forming to full T6 properties by precipitation heat treatment. Uses include architectural applications, bicycle frames, transportation equipment, bridge railings, and welded structures.

**7xxx Series.** Zinc, in amounts of 1 to 8%, is the major alloying element in 7xxx series alloys, and when coupled with a smaller percentage of magnesium it results in heat-treatable alloys of moderate to very high strength. Usually other elements, such as copper and chromium, are also added in small quantities. 7xxx series alloys are

ATTACHMENT II - ASM Specialty Handbook, Aluminum and Aluminum Alloys, J. Davis, ed., p.p. 29-30 and 59-62, ASM International (1993))

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Table 1 Comparative corrosion and fabrication characteristics and typical applications of wrought aluminum alloys

Alloy group	Resistance to corrosion		Weldability		Formability		Resistance to stress corrosion cracking		See typical applications of alloys
	General	Stress corrosion cracking	Weldability	Mechanical	General	Formability	Resistance to stress corrosion cracking	Resistance to stress corrosion cracking	
5xxx series	A	A	A	A	A	A	A	A	Chemical equipment, railroad tank cars
5052	A	A	A	A	A	A	A	A	
5083	A	A	A	A	A	A	A	A	
5154	A	A	A	A	A	A	A	A	
5182	A	A	A	A	A	A	A	A	
5253	A	A	A	A	A	A	A	A	
5356	A	A	A	A	A	A	A	A	
5456	A	A	A	A	A	A	A	A	
5652	A	A	A	A	A	A	A	A	
5754	A	A	A	A	A	A	A	A	
5952	A	A	A	A	A	A	A	A	
6xxx series	A	A	A	A	A	A	A	A	Sheet-metal work, spun hollowware, flat stock
6061	A	A	A	A	A	A	A	A	
6063	A	A	A	A	A	A	A	A	
6105	A	A	A	A	A	A	A	A	
6262	A	A	A	A	A	A	A	A	
6350	A	A	A	A	A	A	A	A	
6463	A	A	A	A	A	A	A	A	
6603	A	A	A	A	A	A	A	A	
6803	A	A	A	A	A	A	A	A	
6903	A	A	A	A	A	A	A	A	
7xxx series	A	A	A	A	A	A	A	A	Electrolytic capacitor foil, chemical equipment, railroad tank cars
7050	A	A	A	A	A	A	A	A	
7055	A	A	A	A	A	A	A	A	
7060	A	A	A	A	A	A	A	A	
7075	A	A	A	A	A	A	A	A	
7150	A	A	A	A	A	A	A	A	
7250	A	A	A	A	A	A	A	A	
7350	A	A	A	A	A	A	A	A	
7450	A	A	A	A	A	A	A	A	
7550	A	A	A	A	A	A	A	A	
7650	A	A	A	A	A	A	A	A	
7750	A	A	A	A	A	A	A	A	
7850	A	A	A	A	A	A	A	A	
7950	A	A	A	A	A	A	A	A	
8xxx series	A	A	A	A	A	A	A	A	Electrolytic capacitors
8000	A	A	A	A	A	A	A	A	
8001	A	A	A	A	A	A	A	A	
8002	A	A	A	A	A	A	A	A	
8003	A	A	A	A	A	A	A	A	
8004	A	A	A	A	A	A	A	A	
8005	A	A	A	A	A	A	A	A	
8006	A	A	A	A	A	A	A	A	
8007	A	A	A	A	A	A	A	A	
8008	A	A	A	A	A	A	A	A	
8009	A	A	A	A	A	A	A	A	
8010	A	A	A	A	A	A	A	A	
8011	A	A	A	A	A	A	A	A	
8012	A	A	A	A	A	A	A	A	
8013	A	A	A	A	A	A	A	A	
8014	A	A	A	A	A	A	A	A	
8015	A	A	A	A	A	A	A	A	
8016	A	A	A	A	A	A	A	A	
8017	A	A	A	A	A	A	A	A	
8018	A	A	A	A	A	A	A	A	
8019	A	A	A	A	A	A	A	A	
8020	A	A	A	A	A	A	A	A	
8021	A	A	A	A	A	A	A	A	
8022	A	A	A	A	A	A	A	A	
8023	A	A	A	A	A	A	A	A	
8024	A	A	A	A	A	A	A	A	
8025	A	A	A	A	A	A	A	A	
8026	A	A	A	A	A	A	A	A	
8027	A	A	A	A	A	A	A	A	
8028	A	A	A	A	A	A	A	A	
8029	A	A	A	A	A	A	A	A	
8030	A	A	A	A	A	A	A	A	
8031	A	A	A	A	A	A	A	A	
8032	A	A	A	A	A	A	A	A	
8033	A	A	A	A	A	A	A	A	
8034	A	A	A	A	A	A	A	A	
8035	A	A	A	A	A	A	A	A	
8036	A	A	A	A	A	A	A	A	
8037	A	A	A	A	A	A	A	A	
8038	A	A	A	A	A	A	A	A	
8039	A	A	A	A	A	A	A	A	
8040	A	A	A	A	A	A	A	A	
8041	A	A	A	A	A	A	A	A	
8042	A	A	A	A	A	A	A	A	
8043	A	A	A	A	A	A	A	A	
8044	A	A	A	A	A	A	A	A	
8045	A	A	A	A	A	A	A	A	
8046	A	A	A	A	A	A	A	A	
8047	A	A	A	A	A	A	A	A	
8048	A	A	A	A	A	A	A	A	
8049	A	A	A	A	A	A	A	A	
8050	A	A	A	A	A	A	A	A	
8051	A	A	A	A	A	A	A	A	
8052	A	A	A	A	A	A	A	A	
8053	A	A	A	A	A	A	A	A	
8054	A	A	A	A	A	A	A	A	
8055	A	A	A	A	A	A	A	A	
8056	A	A	A	A	A	A	A	A	
8057	A	A	A	A	A	A	A	A	
8058	A	A	A	A	A	A	A	A	
8059	A	A	A	A	A	A	A	A	
8060	A	A	A	A	A	A	A	A	
8061	A	A	A	A	A	A	A	A	
8062	A	A	A	A	A	A	A	A	
8063	A	A	A	A	A	A	A	A	
8064	A	A	A	A	A	A	A	A	
8065	A	A	A	A	A	A	A	A	
8066	A	A	A	A	A	A	A	A	
8067	A	A	A	A	A	A	A	A	
8068	A	A	A	A	A	A	A	A	
8069	A	A	A	A	A	A	A	A	
8070	A	A	A	A	A	A	A	A	
8071	A	A	A	A	A	A	A	A	
8072	A	A	A	A	A	A	A	A	
8073	A	A	A	A	A	A	A	A	
8074	A	A	A	A	A	A	A	A	
8075	A	A	A	A	A	A	A	A	
8076	A	A	A	A	A	A	A	A	
8077	A	A	A	A	A	A	A	A	
8078	A	A	A	A	A	A	A	A	
8079	A	A	A	A	A	A	A	A	
8080	A	A	A	A	A	A	A	A	
8081	A	A	A	A	A	A	A	A	
8082	A	A	A	A	A	A	A	A	
8083	A	A	A	A	A	A	A	A	
8084	A	A	A	A	A	A	A	A	
8085	A	A	A	A	A	A	A	A	
8086	A	A	A	A	A	A	A	A	
8087	A	A	A	A	A	A	A	A	
8088	A	A	A	A	A	A	A	A	
8089	A	A	A	A	A	A	A	A	
8090	A	A	A	A	A	A	A	A	
8091	A	A	A	A	A	A	A	A	
8092	A	A	A	A	A	A	A	A	
8093	A	A	A	A	A	A	A	A	
8094	A	A	A	A	A	A	A	A	
8095	A	A	A	A	A	A	A	A	
8096	A	A	A	A	A	A	A	A	
8097	A	A	A	A	A	A	A	A	
8098	A	A	A	A	A	A	A	A	
8099	A	A	A	A	A	A	A	A	
8100	A	A	A	A	A	A	A	A	

(continued)

(a) Ratings: A Through E are relative ratings in decreasing order of merit, based on resistance to sodium chloride solution by immersion testing. Alloys with A and B ratings can be used in saltwater environments without treatment. Rating C is for use in saltwater environments with treatment. Rating D is for use in saltwater environments with treatment. Rating E is for use in saltwater environments with treatment. (b) Stress corrosion cracking: A Through E are relative ratings in decreasing order of merit, based on resistance to stress corrosion cracking by immersion testing. Alloys with A and B ratings can be used in saltwater environments without treatment. Rating C is for use in saltwater environments with treatment. Rating D is for use in saltwater environments with treatment. Rating E is for use in saltwater environments with treatment. (c) Formability: A Through E are relative ratings in decreasing order of merit, based on resistance to formability by immersion testing. Alloys with A and B ratings can be used in saltwater environments without treatment. Rating C is for use in saltwater environments with treatment. Rating D is for use in saltwater environments with treatment. Rating E is for use in saltwater environments with treatment. (d) Weldability: A Through E are relative ratings in decreasing order of merit, based on resistance to weldability by immersion testing. Alloys with A and B ratings can be used in saltwater environments without treatment. Rating C is for use in saltwater environments with treatment. Rating D is for use in saltwater environments with treatment. Rating E is for use in saltwater environments with treatment. (e) Typical applications: A Through E are relative ratings in decreasing order of merit, based on resistance to typical applications by immersion testing. Alloys with A and B ratings can be used in saltwater environments without treatment. Rating C is for use in saltwater environments with treatment. Rating D is for use in saltwater environments with treatment. Rating E is for use in saltwater environments with treatment.







**WORKING METHOD FOR ALUMINUM ALLOY EXTRUDED SHAPE**

Publication number: JP2000178704 (A)

Also published as:

Publication date: 2000-08-27

☐ JP3504823 (B2)

Inventor(s): TANIGAWA HISAO; OHORI KOICHI

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Classification:

- International: C22F1/00; C22C21/02; C22C21/06; C22F1/05; C22F1/00; C22F1/06; C22C21/02; C22C21/06; C22F1/05; C22F1/00; (IPC1-7): C22F1/00; C22F1/05; C22C21/02

- European:

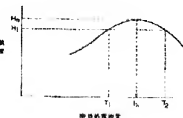
Application number: JP19980353639 19981211

Priority number(s): JP19980353639 19981211

Abstract of JP 2000178704 (A)

**PROBLEM TO BE SOLVED:** To provide a working method for an Al alloy extruded shape, capable of manufacturing an Al alloy extruded shape excellent in bendability and energy absorption characteristic.

**SOLUTION:** An Al alloy, having a composition consisting of, by weight, 0.6-1.2% Si, 0.5-1.0% Mg, 0.1-0.4% Fe, 0.2-0.6% Mn, 0.005-0.1% Ti, 0.05-0.3% Cr and/or 0.05-0.25% Zr, and the balance Al with inevitable impurities, is extruded. The resultant Al alloy extruded shape is subjected to primary heat treatment under the aging condition T1 before the aging condition Th where the highest strength hth of the Al alloy is reached. Then, bending is applied. Subsequently, secondary heat treatment is carried out under the aging condition T2 beyond the aging condition Th where the highest strength hth of the Al alloy is reached.



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